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Resonant Tunneling Heterostructure Devices – Dependencies on Thickness and Number of Quantum Wells

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Abstract. We present numerical results for GaAs/AlGaAs double-barrier resonant tunneling heterostructure devices. A particular emphasis is given to the influence of quantum well thickness and number of quantum well layers on current-voltage characteristic and carrier density profile. In the paper, we discuss results obtained for spatial dependencies of carrier densities, the peak and the valley current density, and corresponding potentials in N-shaped current-voltage characteristics for various resonant tunneling heterostructures. Results are based on the transient quantum drift-diffusion model. They are obtained by solving a coupled system of partial differential equations directly and, in contrast to previous analysis, no decoupling algorithms, procedures, or methods are used.

1 Introduction

Semiconductor devices that rely on quantum tunneling through potential barriers are playing an increasingly important role in advanced microelectronic applications, including multiple-state logic, memory devices, and high-frequency oscillators [1,2,3]. The local charge accumulation in quantum wells and nonlinear processes of charge transport across the barriers have been found to provide a number of mechanisms for Negative Differential Resistance/Conductance (NDR/C), bistability of the current at a given voltage, and nonlinear dynamics [4]. The N-shape of Current-Voltage Characteristics (IVC) may be adopted for realizing various logic functions.

By controlled layer-by-layer epitaxial growth of heterostructures in combination with lateral patterning, intricate artificial nanostructures with arbitrary shapes of barriers and wells can be designed and fabricated. Such bandstructure engineering can produce novel semiconductor devices with desired transport and optical properties. The aim of the current research activity in the field is not only to understand the complex and sometimes chaotic spatio-temporal dynamics of charge carriers in such structures, but also to make efficient use of those nonlinear transport properties in specific switching and oscillating electronic devices [4].

Physical device models are based on the physics of carrier transport, and can provide a great insight into the detailed operation of the device. In what follows, particular emphasis is placed upon low-dimensional GaAs/AlGaAs structures and the

nonlinear feedback between the space charges and the transport processes inherent in such structures. In Sect. 2, we present the physical model, as well as time and space discretization. Geometry and relevant parameters of the devices are mainly given in Sect. 3. A short description of numerical simulation is given in Sect. 4. Obtained results are presented and discussed in Sect. 5.

2 Theory and the Model

2.1 Origin and Validity of the Model

In the present paper, we employ the Transient Quantum Drift-Diffusion Model (TQDDM). It is a first-moment version of the isothermal Transient Quantum Hydrodynamic Model (TQHDM), where the velocity convection term is neglected [5]. The origin of the model is due to Ancona *et al.* [5,6]. The equation of state for the electron gas is generalized and density-gradient dependencies are included, which allow to account for quantum effects [5,6].

The TQDDM is limited to “high” temperatures ($T_0 \geq 77$ K) and “low” electron densities ($n \leq 3 \cdot 10^{19}$ electrons/cm³), conditions often satisfied in semiconductor structure applications of most interest [7]. The density-gradient expansion is only valid if the coefficient ε , related to the quantum-correction term, is very small, $\varepsilon \ll 1$ [7]. Thus, the lowest-order density-gradient theory gives the best results when the characteristic length is large ($L > 10$ nm), and the effective mass close to its free-electron value [7].

2.2 The Transient Quantum Drift-Diffusion Model

A detailed derivation of the TQDDM is given by Pinnau *et al.* [8,9]. Basically, diffusion scaling is introduced in the TQHDM and the TQDDM is derived from a zero relaxation time limit.

The scaled TQDDM equations in 1D, stated on a bounded domain $\Omega \in R$, read [9]

$$n_t = (nF_x)_x, \quad (1a)$$

$$-\varepsilon^2 \frac{1}{\sqrt{n}} (\sqrt{n})_{xx} + \log(n) + V = F, \quad (1b)$$

$$-\lambda^2 V_{xx} = n - C, \quad (1c)$$

where the dependent variables are: the electron density n , the quantum quasi-Fermi level F , and the electrostatic potential V . The time-independent doping profile C represents the distribution of charged background ions. The scaled Planck constant ε , the scaled Debye length λ , and the scaled relaxation time τ_0 (used in the numerical time discretization) are defined as

$$\varepsilon^2 = \frac{\hbar^2}{6m_e^*k_B T_0 L^2}, \quad \lambda^2 = \frac{\varepsilon_s k_B T_0}{q^2 C_m L^2}, \quad \tau_0^2 = \frac{k_B T_0 \tau^2}{m_e^* L^2},$$

where physical constants are: the reduced Planck constant \hbar , the Boltzmann constant k_B , and the elementary charge q . Physical parameters are: the effective electron mass m_e^* , the device operating temperature T_0 , the permittivity ε_s , and the relaxation time τ , which depend on the material and the operating conditions of the device. The maximum absolute value of the doping profile C is denoted as C_m , and L is the characteristic (device) length.

The first term on the LHS of eq. (1b) is so-called quantum Bohm potential. The scaled current density, according to eq. (1a), is given by the following expression:

$$J = -nF_x. \quad (2)$$

Now, it is possible to introduce an external potential, modeling discontinuities in the conduction band, which occur in the resonant tunneling structures, and other semiconductor heterostructure devices [10]. For that reason, one must replace, in (1b), the potential V by $V + B$ [10], where B is a step function representing the nonnegative quantum well potential. The maximum value B_m , of the step-function B , depends on the content of Al in the ternary alloy composition. It is assumed that $B_m = 0.4$ eV, which corresponds to 65% Al in AlGaAs at 300 K [2].

In order to get a well-posed problem, the system of eqs. (1) has to be supplemented with appropriate boundary and initial conditions.

The electron density is assumed to fulfill local charge neutrality at the Ohmic contacts. Further, it is natural to assume that there is no normal component of the total current (including the quantum current) along the insulating part of the boundary. Finally, we require that no quantum effects occur at the contacts. These boundary conditions are physically motivated and commonly employed in quantum semiconductor modeling [10]. The numerical investigations in [11] underline the reasonability of this choice.

The boundaries are assumed to be at grid points 0 and M (where $M + 1$ is the total number of grid points), e.g., at positions $x = 0$ and $x = L$, or in scaled coordinates, $x = 0$ and $x = 1$, respectively. The corresponding boundary conditions are given below:

$$\rho_{k,0} = \sqrt{C_0}, \quad F_{k,0} = 0, \quad V_{k,0} = 0, \quad (3)$$

$$\rho_{k,M} = \sqrt{C_M}, \quad F_{k,M} = U, \quad V_{k,M} = U, \quad (4)$$

where a new variable $\rho = \sqrt{n}$ has been introduced, and U is an applied voltage.

The initial conditions for ρ , F and V are required to start numerical simulation in equilibrium case ($U = 0$ V). It is natural to set the initial values for F and V to zero, while one has the full freedom to choose/guess the initial value for ρ .

2.3 Discretization in Time and Space

One of the main requirements for the time-discretization is that the scheme should be stable [12,13]. Further, there is no need for higher-order schemes, since the overall discretization error will be dominated by the one introduced through the space-discretization [14,15]. Thus, schemes with first-order accuracy in the time step are sufficient. Moreover, in most of the numerical simulations for the classical Drift-Diffusion Model (DDM), schemes based on backward time differences are employed [12,14]. Since the TQDDM is an $O(\xi^2)$ correction of DDM, it is reasonable to assume this to be true [9]. The most prominent scheme fulfilling the above requirements is the implicit backward Euler method [14,15]. Afterwards, a convenient realizable discrete scheme is derived in two steps: a uniform spatial grid ($\Delta x = L/M$) is introduced, and the finite difference method is used [13,14,15].

3 Geometry and Relevant Parameters

A Quantum Well (QW) is a synthetic heterostructure containing a very thin layer (thickness of a few nanometers) of one semiconductor sandwiched between two (thin) layers of another semiconductor with a larger bandgap [3]. A superlattice is another important nanostructure, which arises if several alternating layers of two materials with different bandgaps are grown one by one. The potential profile of quantum wells and barriers, which show periodicity, is intimately connected with the charge transport properties of the nanostructures [4].

In the present paper, the basic Double Barrier Resonant Tunneling Diode (DBRTD) consists of a quantum well GaAs layer sandwiched between two AlGaAs layers, each 5 nm thick. This resonant structure is itself sandwiched between two spacer GaAs layers of 5 nm thicknesses and supplemented with two contact GaAs regions, each 25 nm thick. The basic $n^+n^-n^+$ DBRTD in 2D is shown in Fig. 1. A superlattice has a similar structure with more than one barrier-QW-barrier alternating layers sandwiched between the spacer layers and supported with two contact regions.

The contact regions are highly doped (n^+ type) with $C_m = 10^{24} \text{ m}^{-3}$, while the channel is moderately doped (n type) with $C_m = 10^{21} \text{ m}^{-3}$. The distribution of charged background ions is described by the doping profile C , which is time independent. Such a device exhibits NDR/C due to electron tunneling through the potential barriers. A typical stationary N-shape IVC is well known from the literature.

The domain in 1D case is the interval $\Omega = [0, L]$, $L > 0$ being the device length. The device length is a sum of all layers/regions of the heterostructure. The relaxation time is fixed at $\tau = 10^{-12} \text{ s}$ [10]. It is assumed that the devices are operating at liquid-nitrogen temperature, $T_0 = 77 \text{ K}$. The effective electron mass is chosen to be $m_e^* = 0.067 \cdot m_0$, where m_0 is electron rest mass. The permittivity of GaAs/AlGaAs is chosen as $\epsilon_s = 13.1 \cdot \epsilon_0$, where ϵ_0 is permittivity in vacuum. It is also assumed that we operate close to the thermal equilibrium.

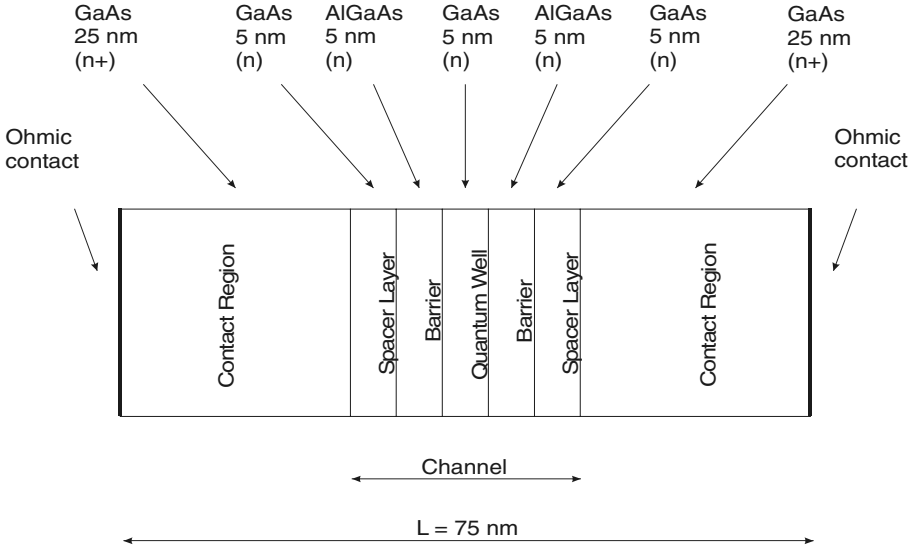


Fig. 1. Basic double barrier resonant tunneling semiconductor structure

4 Numerical Simulation

The numerical simulation is implemented in the Matlab. The system of discretized equations is solved in fully coupled manner for the first time, i.e., in the present work no decoupling algorithms, procedures, or methods are used.

To solve the nonlinear discrete system of equations, which follows from (1), the Newton Iteration Procedure (NIP) is employed [16], where the previous time step is used as initial guess. The following termination criterion is used to stop NIP:

$$\max_{k,l} \left(\left| u_{k,l}^{new} - u_{k,l}^{old} \right| \right) \leq \tau_0 10^{-6}, \quad (5)$$

where u denotes one of the variables ρ , F and V . The indexes k and l correspond to the time and the space discretization, respectively. Convergence is assumed when the residuals are smaller than a set tolerance.

The scaled time step is changed during the time evolution. Initially, it is set to 10^{-6} and afterwards increased. The algorithm for the time step is based on two criterions: to speed up the time evolution and to ensure the convergence. The maximum number of required steps in the NIP is less than or equal to 5. Stationary solutions are reached after approximately 150 time steps, depending upon the semiconductor heterostructure under consideration. As a test example for the steady state, the total time is fixed to $T = 100\tau$ and the same results are obtained.

A uniform grid is used for space discretization. The resolution is set to 4 ppnm (points per nm). It gives us $M = 300$ for the basic case of a 75 nm DBRTD. We have checked the convergence of calculated variables (ρ , F , and V) by increasing the resolution (decreasing the grid size) and obtained a relative error less than 1% when 4 ppnm is used.

The obtained results for electron density n , quantum quasi-Fermi level F , and electrostatic potential V are smooth and good agreement in the case of a basic DBRTD heterostructure, also considered in [10], is obtained. Excellent symmetry for equilibrium state ($U = 0$ V) is present for all semiconductor heterostructures under consideration. The current density in steady state has almost the same value at the Ohmic contacts, as required (relative error is less than 0.5%).

5 Results

Here, we present results for different resonant tunneling heterostructures, obtained with the numerical simulation. In the first case, a DBRTD with one QW is considered corresponding to three different QW sizes (2, 5, and 10 nm), while in the second case, superlattices with 2, 3, and 5 QWs are considered, where each QW is of the same size (5 nm). In all presented cases, width of the barriers and space layers are 5 nm. The length of all DBRTDs is fixed to $L = 75$ nm (contact regions are 26.5, 25, and 22.5 nm, respectively), while the length of the superlattices is a sum of all layers (contact regions are fixed to 25 nm). The external potential U is always applied on the RHS Ohmic contact, while the LHS contact is grounded. In what follows, we are interested in the spatial dependency of electron densities and afterwards we comment on obtained current-voltage (IV) characteristics.

The results for electron densities for all three DBRTD (QW width 2, 5, and 10 nm), are given in Fig. 2. As a reference, doping profiles are also given. The applied potential is $U = 0.2$ V, which approximately corresponds to the peak voltage for these three cases. The results are qualitatively different for various thicknesses of the QW. For the case of a 2 nm QW, accumulation of electrons in the QW is evident, however, the resulting density for the “peak”-applied voltage is much less than the doping concentration. For the cases of 5 and 10 nm QW, the accumulation effect of electrons in the QW is indeed very important, and electron densities, for the peak-applied voltage, are significantly larger than doping concentration (two orders of magnitude). This indicates that a minimum thickness of the QW exists, which allows sufficient accumulation of electrons. In addition, the electron density inside the QW shows an increase of spatial asymmetry as the thickness of the QW is increased. A significant electron density reduction is apparent in the barrier regions and the minimums are not the same on both sides of the QW. The densities inside barriers have larger values on the side where the external potential is applied, in contrast to the densities inside the QW. The minimum of the density in the barrier changes significantly with increasing thickness of the QW. In contrast, the peak of the electron density inside QWs of different thicknesses shows “saturation”, i.e., they are almost the same (if the thickness of the QW is large enough, $L_{\text{QW}} \geq 4\text{--}5$ nm).

The results for electron densities for the superlattice heterostructures (with 2, 3, and 5 QWs), are given in Fig. 3. As a reference, doping profiles are also given. The applied potentials are $U = 0.30$ V, 0.35 V, and 0.50 V, which approximately correspond to the peak voltages, respectively. The superlattice heterostructures have different peaks of electron density in different QWs, under the influence of an external potential. In the case of a superlattice with 2 QWs, the charge accumulation in the QW being closer to the side where the external potential is applied, is much smaller, as is the peak. In the case of a superlattice with more than 2 QWs, the peak of

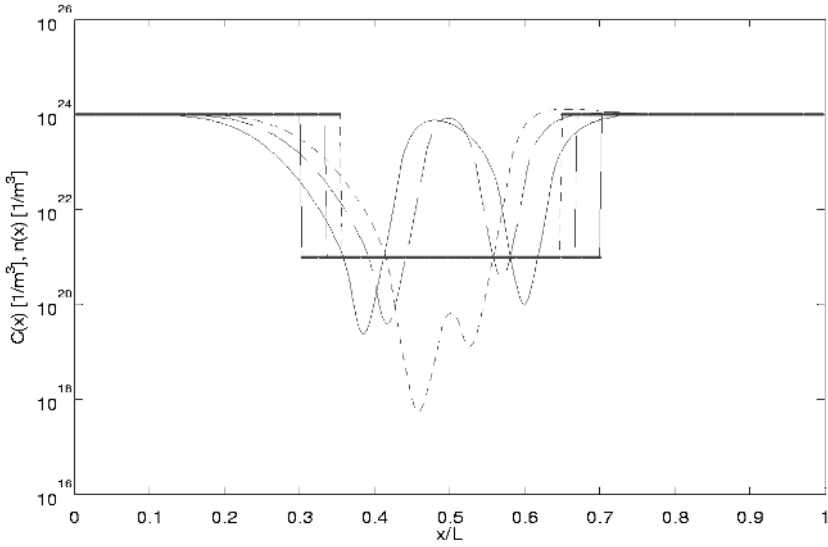


Fig. 2. Densities of electrons $n(x)$ for the DBRTDs with 2, 5, and 10 nm QW; dotted, dashed, and solid curve, respectively; $U = 0.2$ V; corresponding doping profiles $C(x)$ are represented by bold step-curves with the same line coding as for $n(x)$

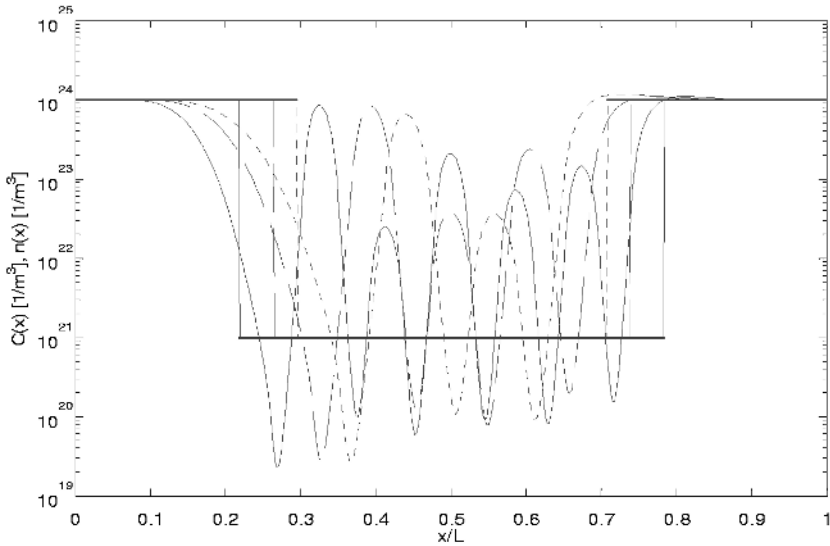


Fig. 3. Densities of electrons $n(x)$ for the superlattices with 2, 3, and 5 QWs; dotted, dashed, and solid curve, respectively; $U = 0.30$ V, 0.35 V, and 0.50 V, respectively; corresponding doping profiles $C(x)$ are represented by bold step-curves with the same line coding as for $n(x)$

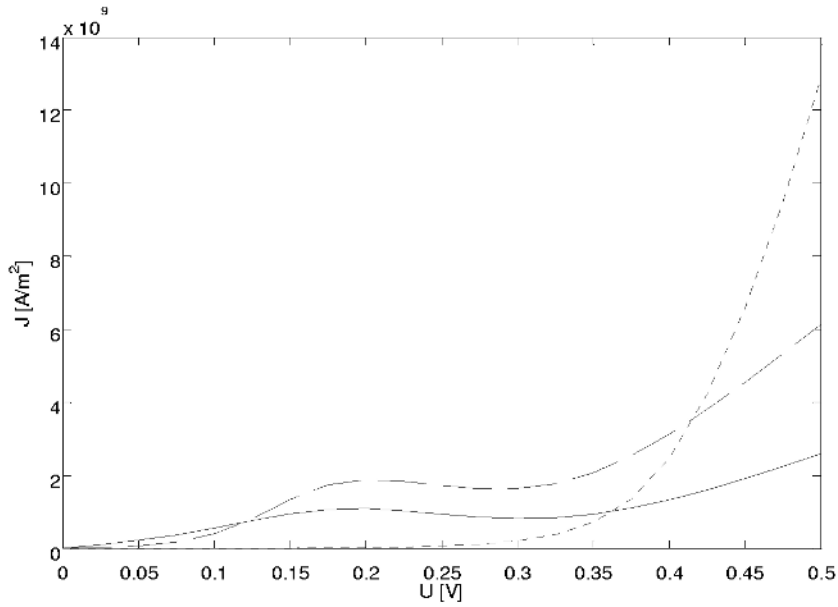


Fig. 4. The IV characteristics for the DBRTDs with 2, 5, and 10 nm QW; dotted, dashed, and solid curve, respectively

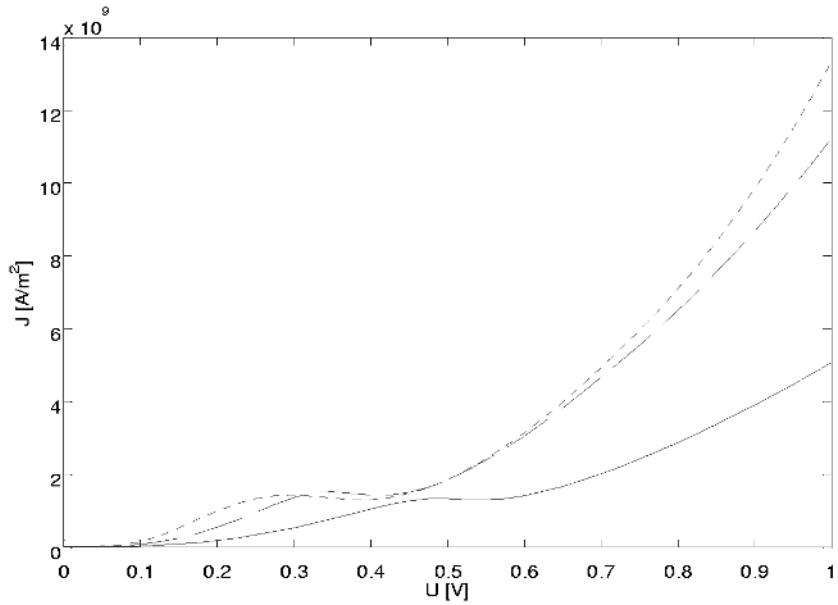


Fig. 5. The IV characteristics for the superlattices with 2, 3, and 5 QWs; dotted, dashed, and solid curve, respectively

the electron density inside QWs changes its magnitude, from the side where the external potential is applied towards the zero-volt (grounded) contact. In general, the QW that is closest to the zero-volt contact has the largest peak of the electron density. In contrast, the minimum of the electron density is always reached in the barrier that is nearby the zero-volt contact.

The IV characteristics for the DBRTDs (QW thickness of 2, 5, and 10 nm), are given in Fig. 4. The IV characteristic for the DBRTD with 2 nm QW does not show NDR/C at all. Changing the width of the QW, a significant change in the IV characteristic occurs, i.e., the current density for the same applied voltage is changed. In addition, the peak and the valley current density, and the relative ratio between them, also change significantly. However, the peak and the valley potential will only slightly differ.

The IV characteristics for the superlattice heterostructures with 2, 3, and 5 QWs (corresponding to $L = 85, 95, 115$ nm, respectively), are given in Fig. 5. It is obvious that an increase of the number of QWs leads to a significant influence on the IV characteristic. At the same time, the peak and the valley potential are also changed, and the current density is reduced. However, the peak and the valley current density will be only slightly changed.

Unfortunately, it is extremely difficult to predict the exact values of the peak and the valley current density in the N-shaped IV characteristics. These values are strongly affected by other mechanisms, such as phonon-assisted tunneling, impurity-assisted tunneling, and scattering. The voltage positions of the current density peaks and valleys are, however, easier to establish, since they are related to energy levels of the subbands.

The computed values of the current density strongly depend on the choice of the intrinsic device parameters. The most important parameters are the effective electron mass and the relaxation time. Thus, the choice of the intrinsic parameters is crucial for the accurate quantitative simulation of the resonant tunneling heterostructures. However, we expect the general tendencies observed here to be correct.

6 Conclusion

The system of fully discretized coupled nonlinear algebraic equations, which follows from (1), is possible to solve avoiding the use of decoupling algorithms, procedures, or methods. The results obtained for different resonant tunneling heterostructures using TQDDM show that the electron density is strongly nonlinear and asymmetric in both, QWs and barriers, under the influence of an external potential. However, in the equilibrium case ($U = 0$ V), perfect symmetry is present. The IV characteristics of the heterostructure nanodevices are quite different, when varying the thickness and the number of QWs of the device. In particular, changing the thickness of the QW, the peak and the valley current density change significantly, while the peak and the valley potential only slightly differ. In contrast, changing the number of the QWs (keeping the size of the QW constant), the peak and the valley potential change dramatically, while the peak and the valley current are almost unchanged.

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Computational Math, Science, and Technology: A New Pedagogical Approach to Math and Science Education	807
<i>Osman Yaşar</i>	

Track on Computer Modeling and Simulation

Resonant Tunneling Heterostructure Devices – Dependencies on Thickness and Number of Quantum Wells	817
<i>Nenad Radulovic, Morten Willatzen, Roderick V.N. Melnik</i>	
Teletraffic Generation of Self-Similar Processes with Arbitrary Marginal Distributions for Simulation: Analysis of Hurst Parameters	827
<i>Hae-Duck J. Jeong, Jong-Suk Ruth Lee, Hyoung-Woo Park</i>	
Design, Analysis, and Optimization of LCD Backlight Unit Using Ray Tracing Simulation	837
<i>Joonsoo Choi, Kwang-Soo Hahn, Heekyung Seo, Seong-Cheol Kim</i>	
An Efficient Parameter Estimation Technique for a Solute Transport Equation in Porous Media	847
<i>Jaemin Ahn, Chung-Ki Cho, Sungkwon Kang, YongHoon Kwon</i>	
HierGen: A Computer Tool for the Generation of Activity-on-the-Node Hierarchical Project Networks	857
<i>Miguel Gutiérrez, Alfonso Durán, David Alegre, Francisco Sastrón</i>	
Macroscopic Treatment to Polymorphic E-mail Based Viruses	867
<i>Cholmin Kim, Soung-uck Lee, Manpyo Hong</i>	
Making Discrete Games	877
<i>Inmaculada García, Ramón Mollá</i>	
Speech Driven Facial Animation Using Chinese Mandarin Pronunciation Rules	886
<i>Mingyu You, Jiajun Bu, Chun Chen, Mingli Song</i>	
Autonomic Protection System Using Adaptive Security Policy	896
<i>Sihn-hye Park, Wonil Kim, Dong-kyoo Kim</i>	
A Novel Method to Support User's Consent in Usage Control for Stable Trust in E-business	906
<i>Gunhee Lee, Wonil Kim, Dong-kyoo Kim</i>	

Track on Financial and Economical Modeling

No Trade under Rational Expectations in Economy (A Multi-modal Logic Approach)	915
<i>Takashi Matsuhisa</i>	